

Heterogeneous microstructural evolution of AA6082 during plastic deformation

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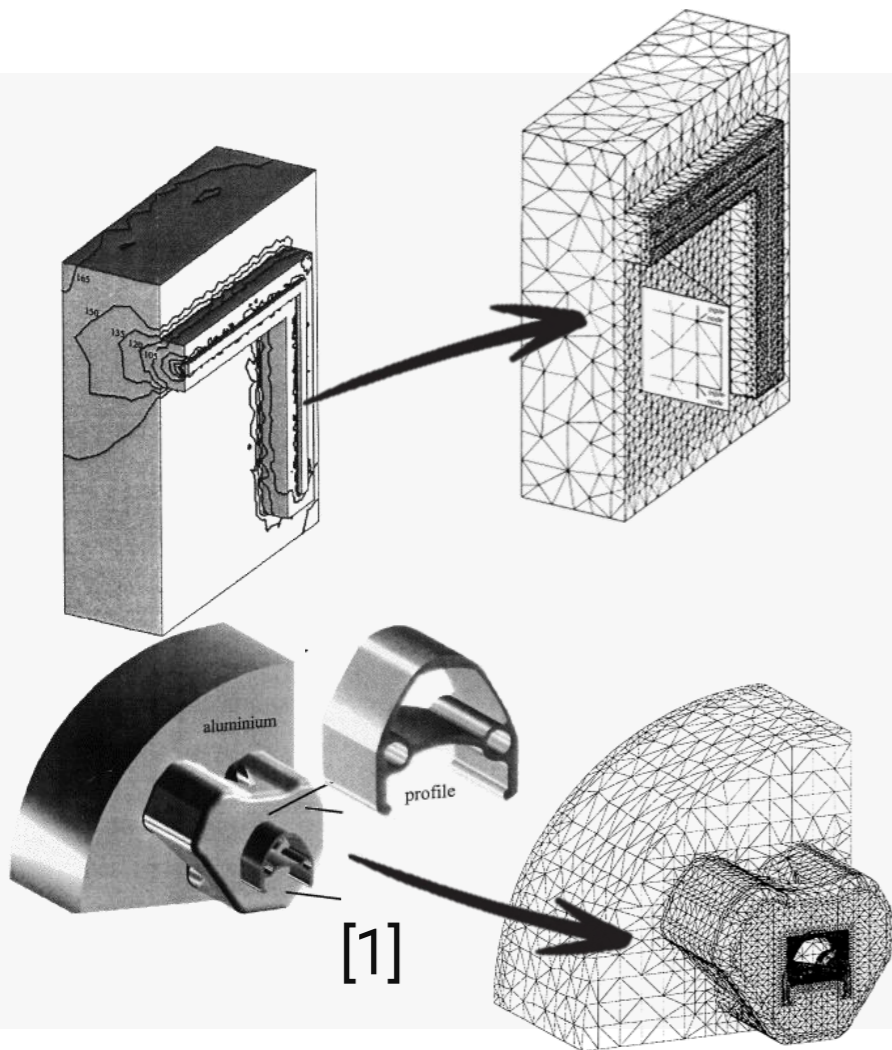
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MOTIVATION



Simulations of industrial processes



Improvement of processing parameters



Desired mechanical properties in final components



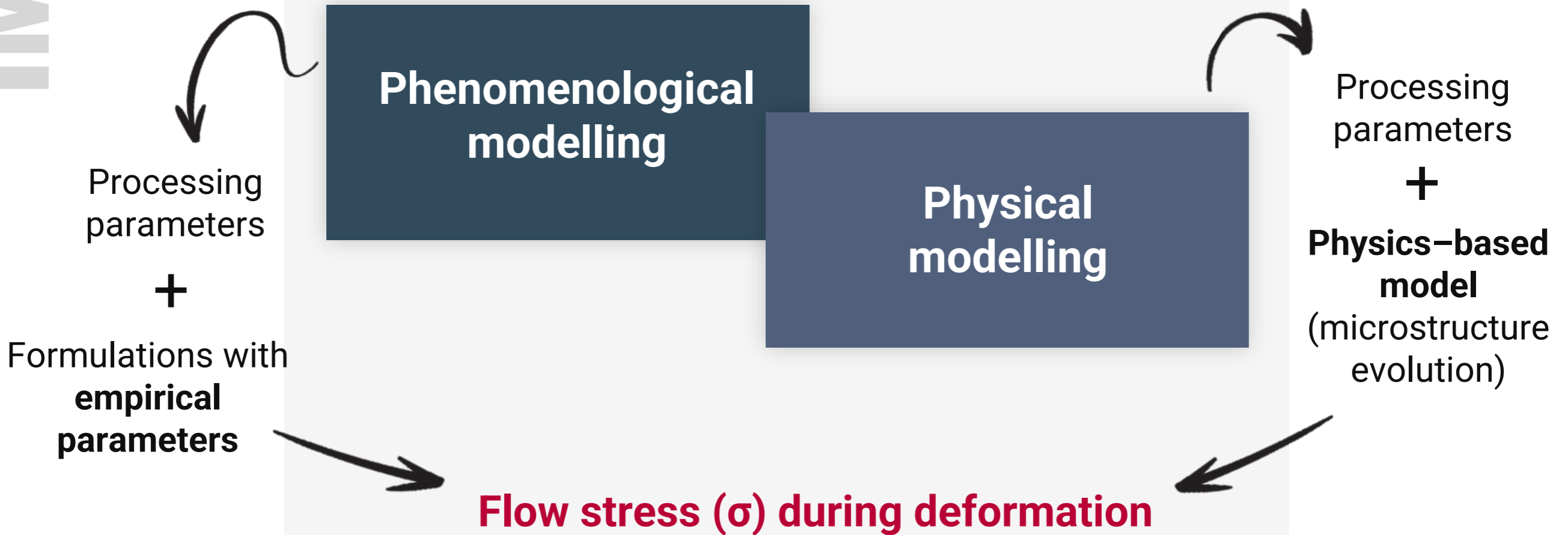
Reduction of costs

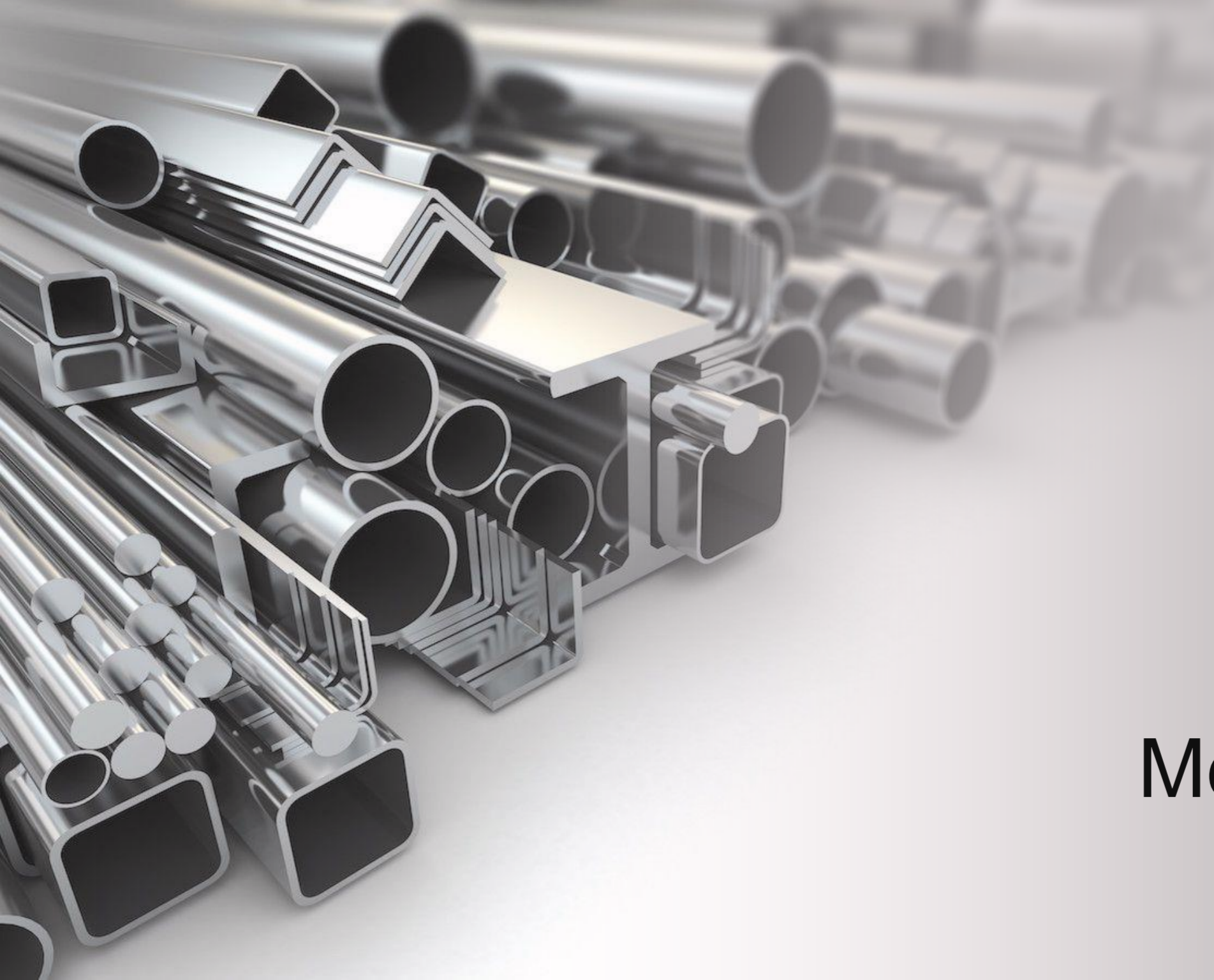
OBJECTIVES

- To study **heterogeneous microstructural evolution** during hot deformation of **AA6082**, representative of **hot extrusion** processes
 - To generate heterogeneous deformation conditions at the lab-scale by compressing **hat-shaped samples**
 - To model and simulate compression tests using a **physics-based model** implemented in **DEFORM** (FEM software)

FOCUS

FEM APPROACHES



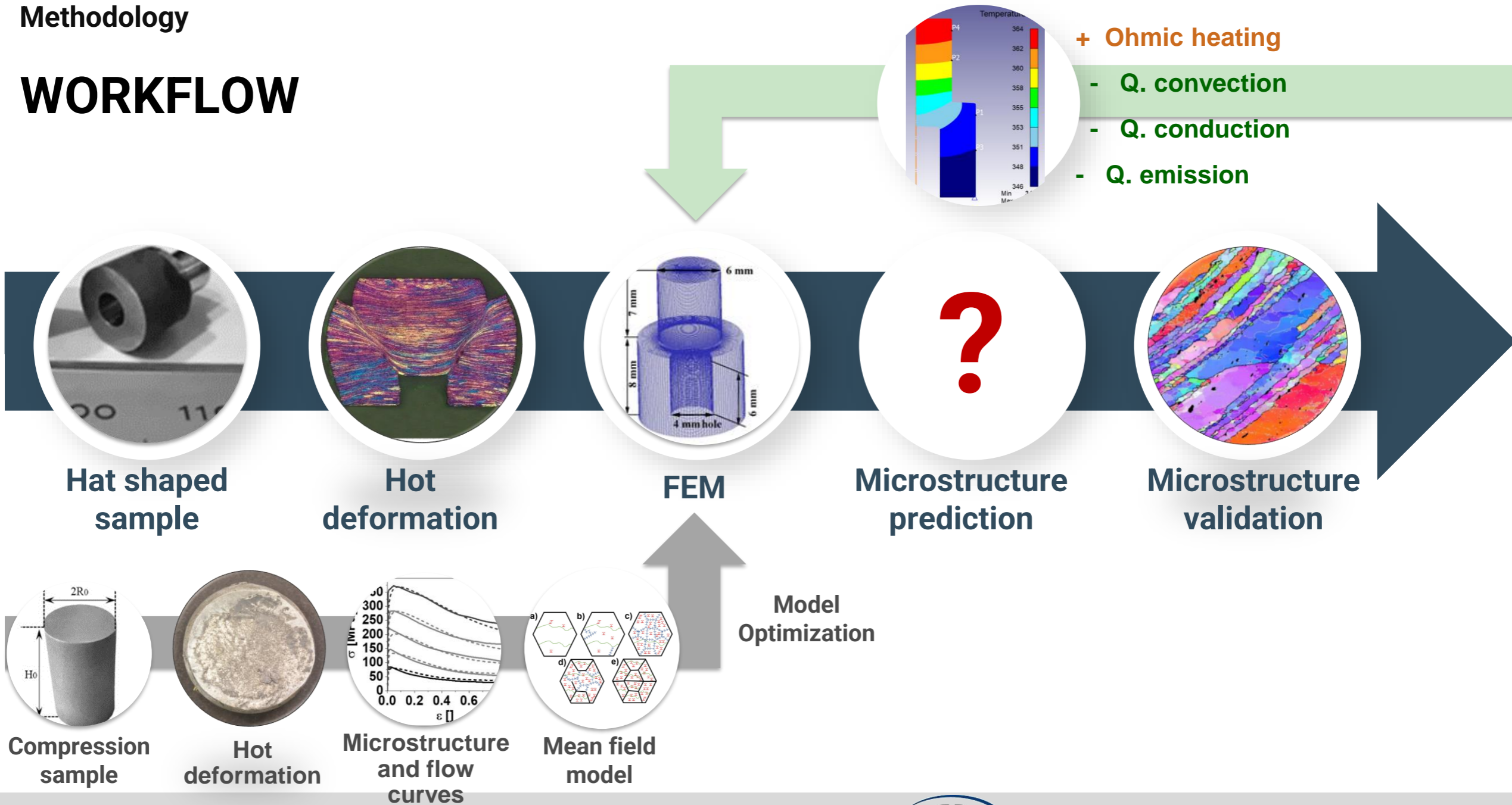


Methodology



Methodology

WORKFLOW



DEFORMATION IN ALUMINIUM

High stacking fault energy material

Dynamic Recovery (DRV)

Cell formation followed by subgrain formation

Continuous Dynamic Recrystallization (CDRX)

Progressive formation and rotation of subgrains into grains

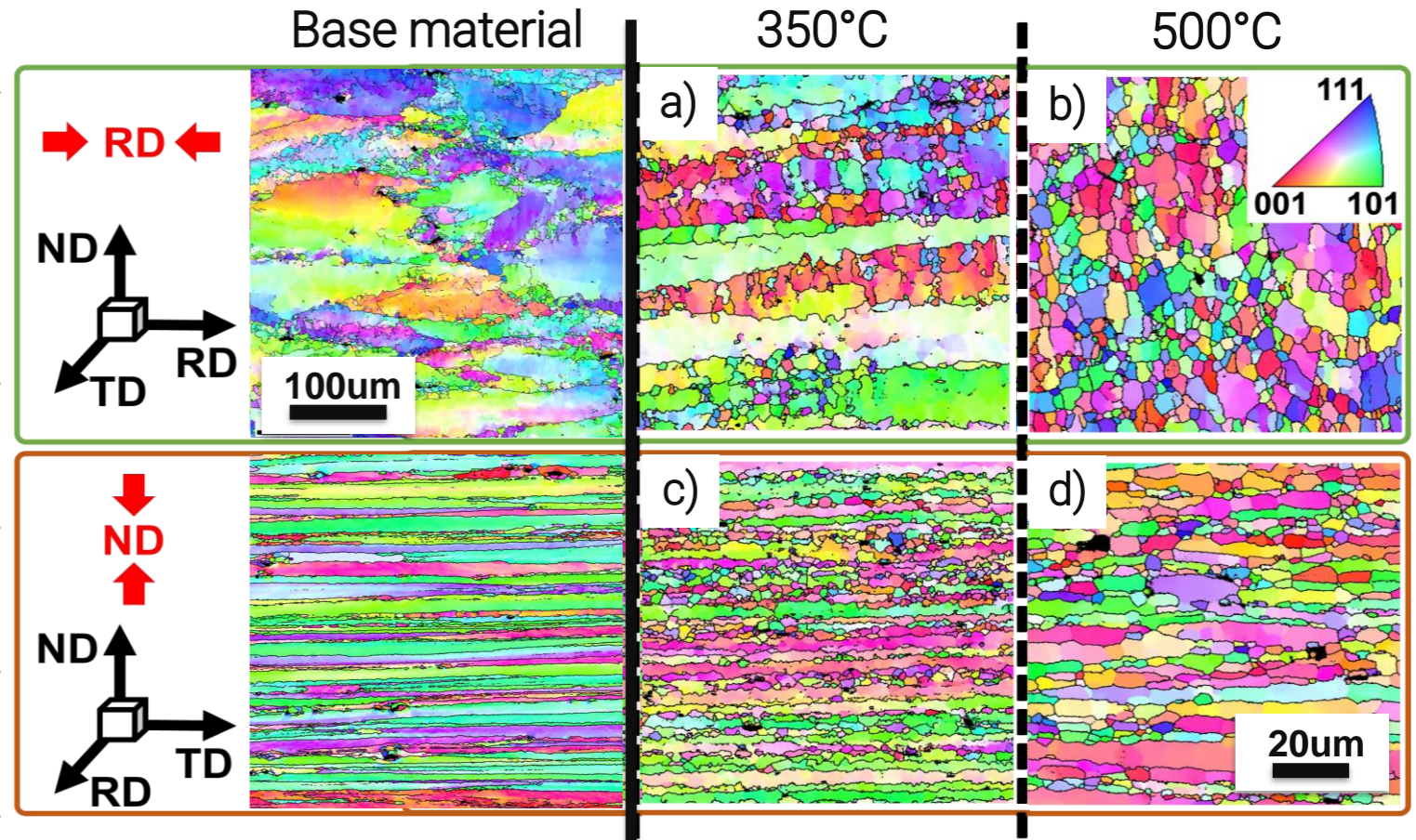
Geometric Dynamic Recrystallization (GDRX)

Migration and pinching off of original HAGB

Intermetallics

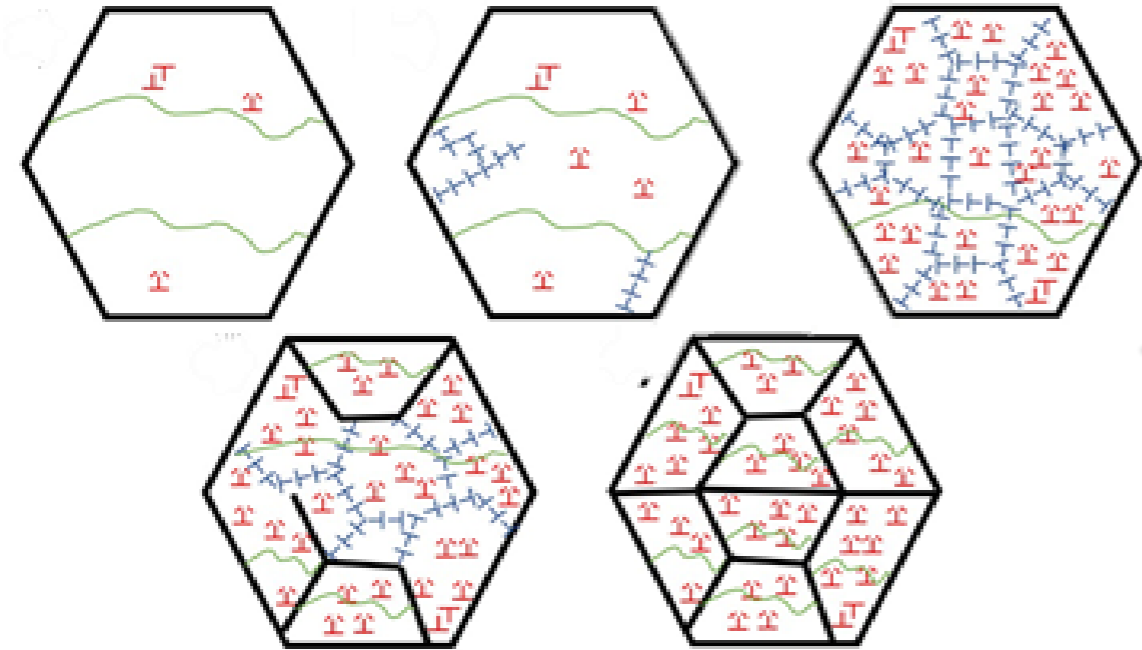
Effect on dislocation motion, pinning of boundaries or PSN

Texture



[2]



MICROSTRUCTURE DESCRIPTION – AA6082



3 densities of dislocations

ρ_w  "Walls"

ρ_m  "Mobile"

ρ_i  "Immobile"  "Dipole"

Fully recrystallized microstructure

Initial stages of deformation

Formation of the substructure

Transformation of LAGBs into HAGBs

Fully recrystallized steady state

CDRX MODEL

Rate equations

$$\sigma_{YS} = \frac{1}{\alpha_{YS}} \ln \left\{ \left(\frac{\dot{\epsilon} \exp\left(\frac{Q_{YS}}{RT}\right)}{A_{YS}} \right)^{\frac{1}{n_{YS}}} + \left[\left(\frac{\dot{\epsilon} \exp\left(\frac{Q_{YS}}{RT}\right)}{A_{YS}} \right)^{\frac{1}{2n_{YS}}} + 1 \right]^{1/2} \right\}$$

Yield stress formulation

$$\sigma = \sigma_{th} + \sigma_{ath}$$

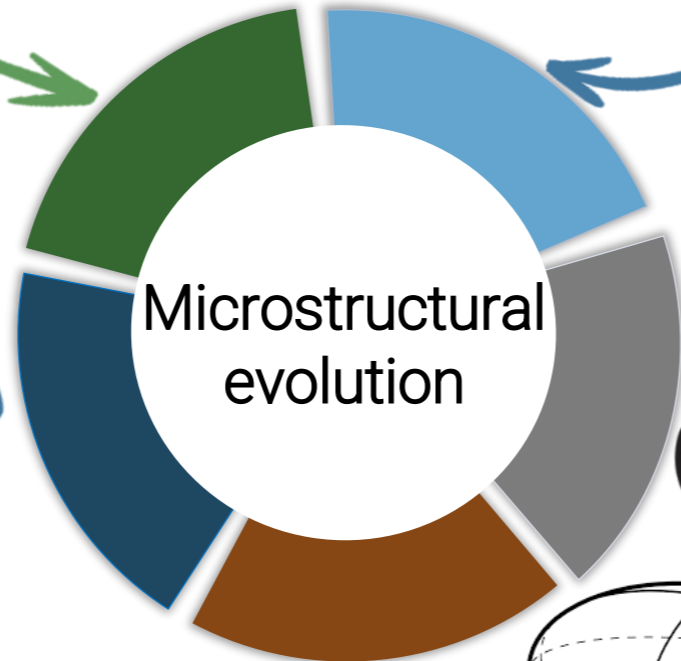
$$\sigma_{th} = \sigma_{YS} - \sigma_{ath}^0$$

$$\sigma_{ath} = abGM \sqrt{\rho_m + \rho_i + F_w \rho_w}$$

Constitutive equations

$$\frac{d\rho_m}{dt} = \rho_m^{3/2} v_{eff} + \frac{\delta_{SG} \rho_m \phi_{sg} v_{eff}}{\lambda^2} - \frac{\rho_m v_{eff}}{\phi_{sg}} - 8 \frac{\rho_m v_c}{\lambda} - \delta_{DRV} \rho_m (\rho_m + \rho_i) v_{eff} - f_{sv} S_{Interface} v_{eff} \rho_m - \frac{2 \rho_m f_{HAGB} v_{HAGB}}{\phi_{sg}}$$

$$\frac{d\rho_i}{dt} = \frac{\rho_m v_{eff}}{\phi_{sg}} - 8 \frac{\rho_i v_c}{\lambda} - \delta_{DRV} \rho_m \rho_i v_{eff} - \frac{2 \rho_m f_{HAGB} v_{HAGB}}{\phi_{sg}}$$



CDRX

$$\frac{dS_v}{dt} = \alpha_{CDRX} \frac{b}{n\theta_0} \Delta\rho_{CDRX} \dot{\epsilon} p - S_v^2 f_{HAGB} v_{HAGB}$$

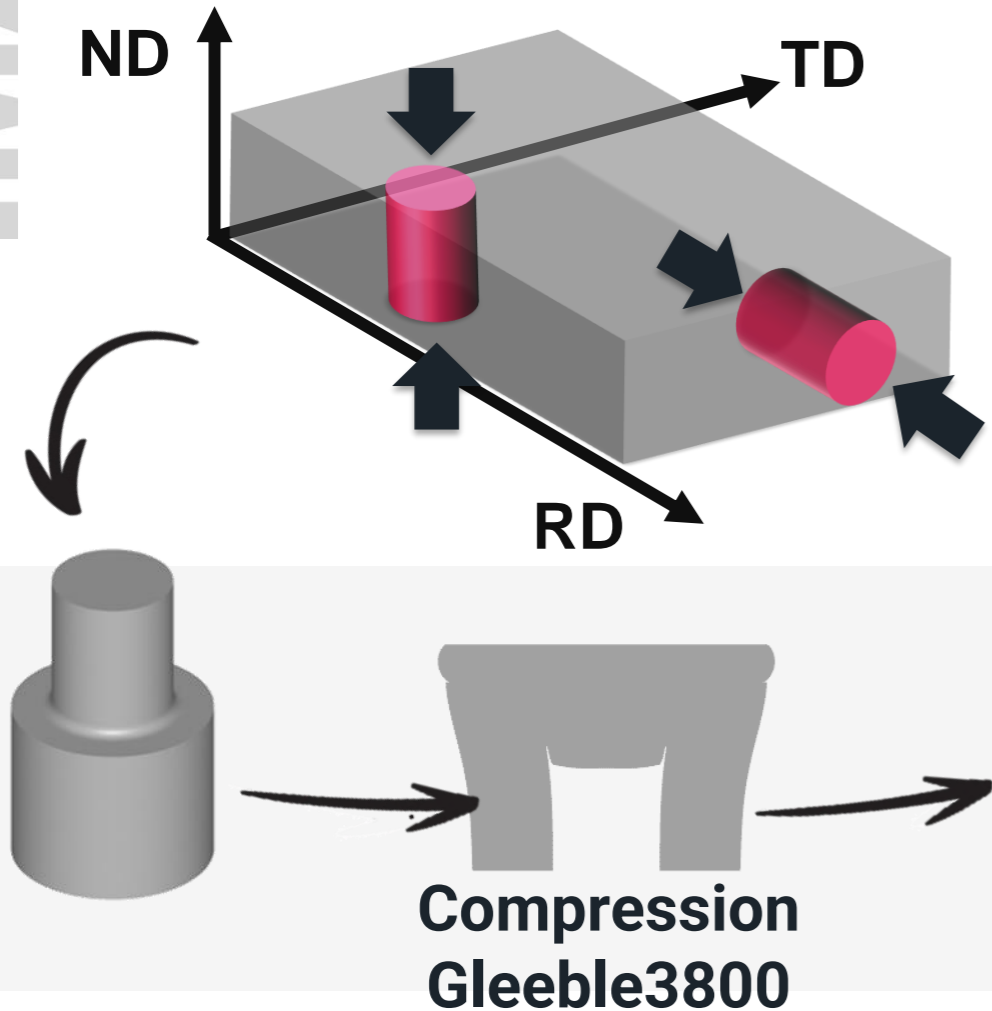
$$\frac{d\bar{\theta}_R}{dt} = (1 - \alpha_{CDRX}) \frac{b}{nS_v} \Delta\rho_{CDRX} \dot{\epsilon} p$$

$$\frac{dl_i}{dt} \text{ for } i = x, z, y$$

[2,3,4]

Methodology

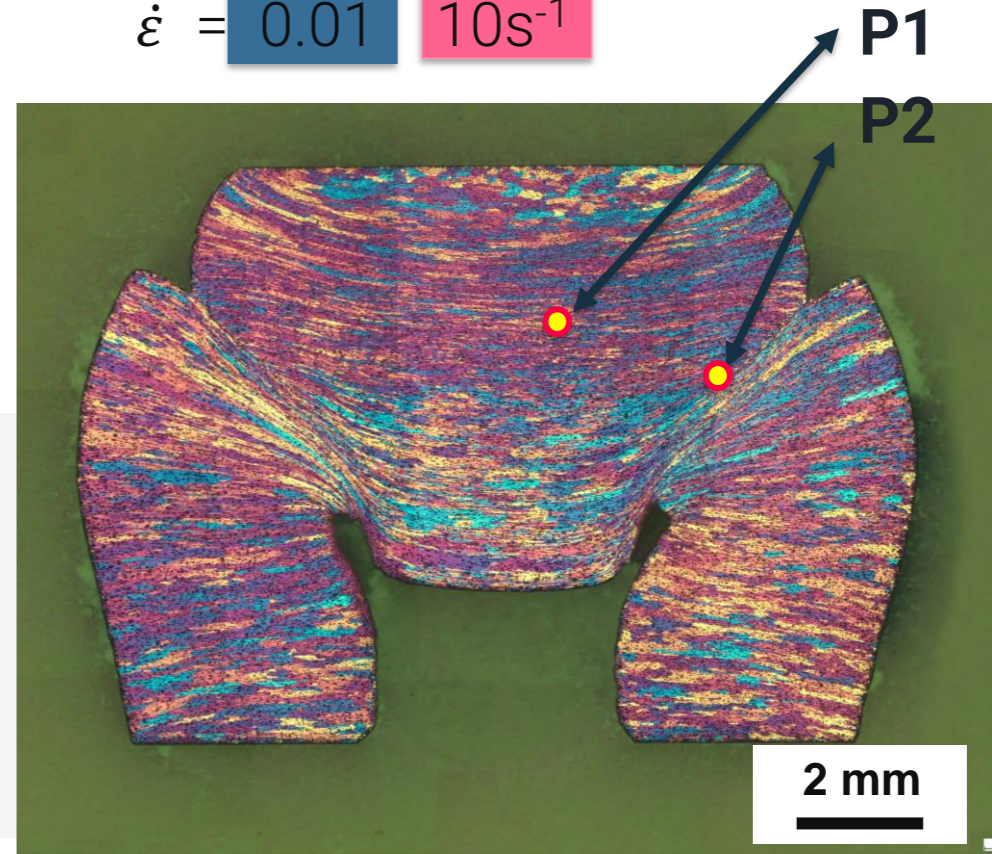
HAT-SHAPED SAMPLES



LOM
EBSD

Tested conditions:

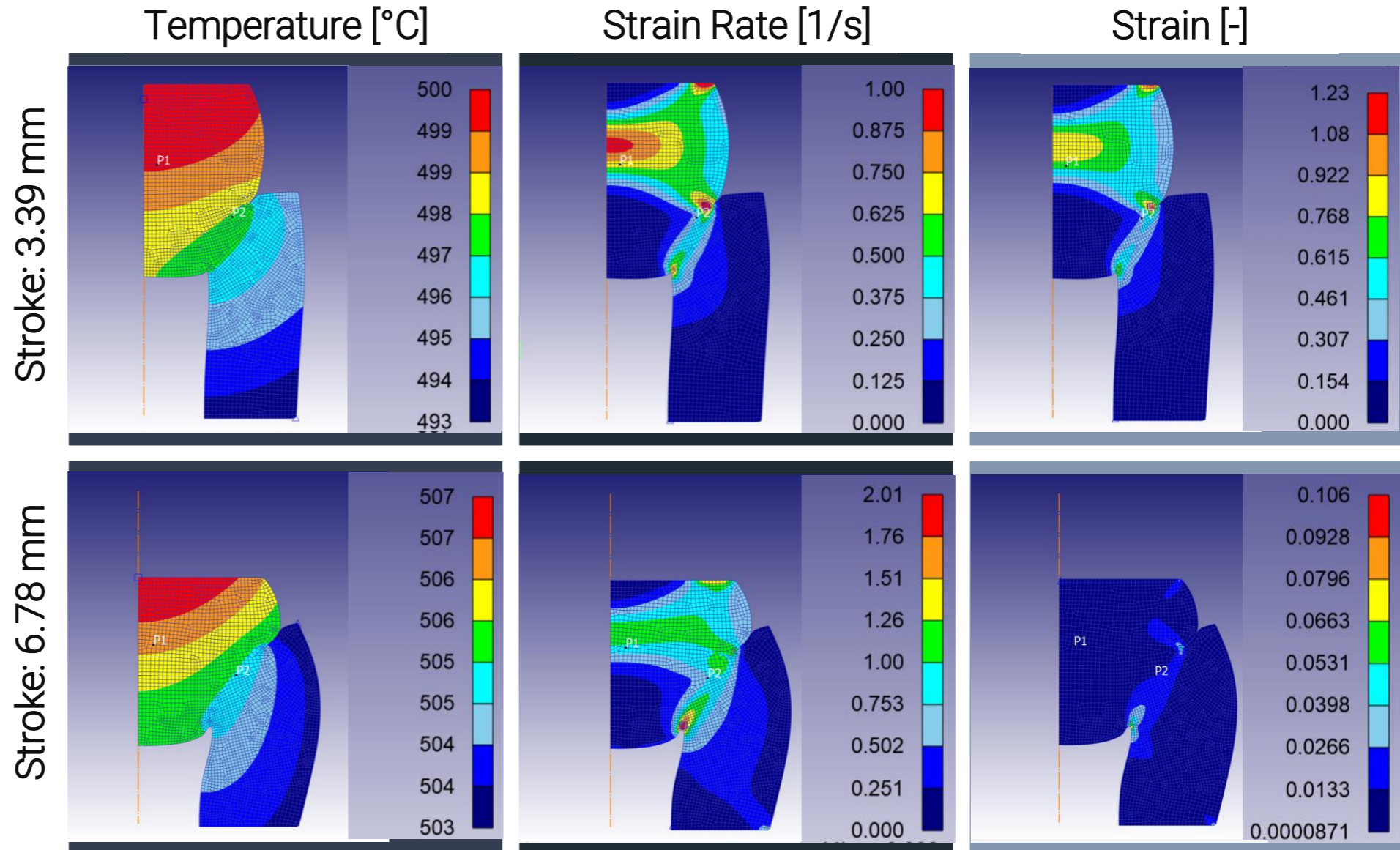
$T =$	350	500°C
$\dot{\epsilon} =$	0.01	10s ⁻¹



HAT-SHAPED SAMPLES

Why is this geometry interesting?

- Flow curves?
- Local T , ϵ , $\dot{\epsilon}$ gradients
- Representative of **HOT EXTRUSION**





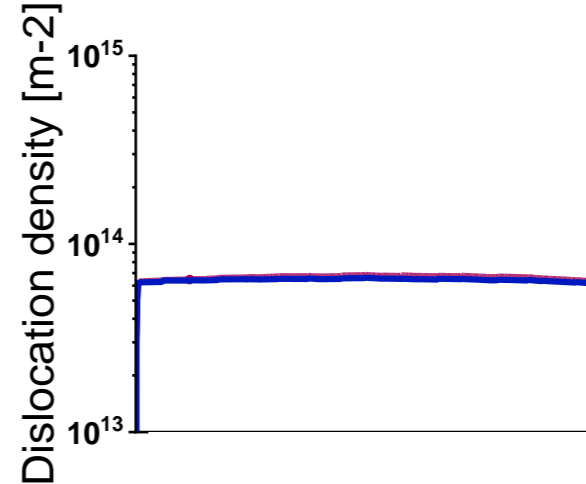
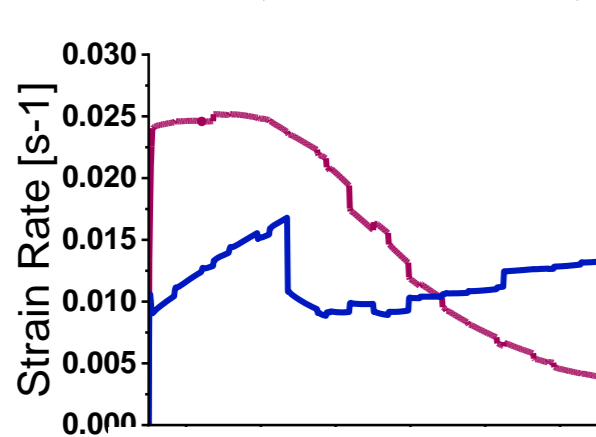
Selected results



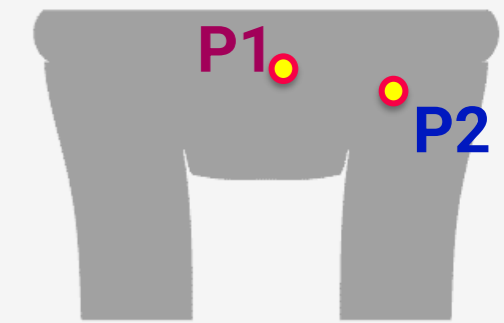
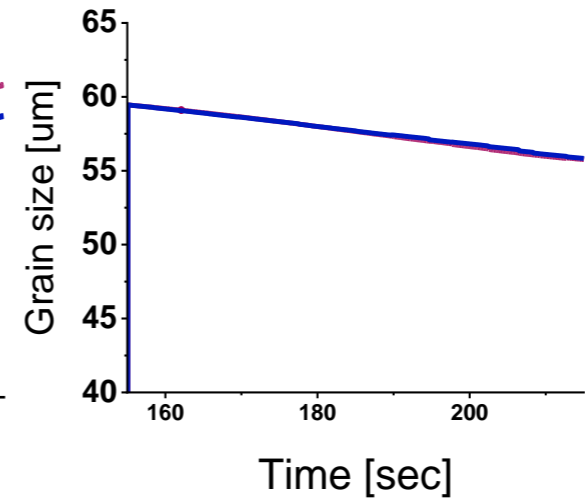
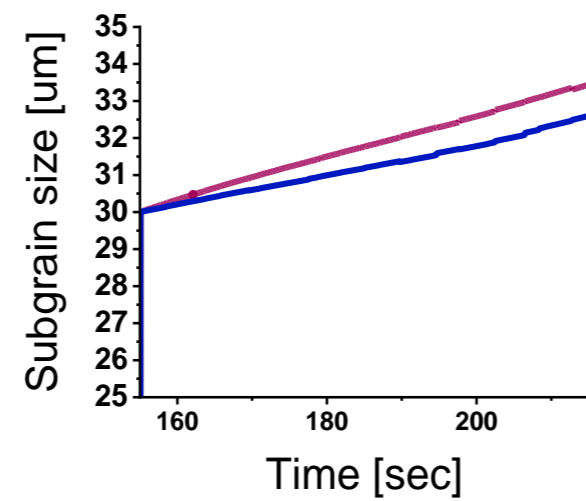
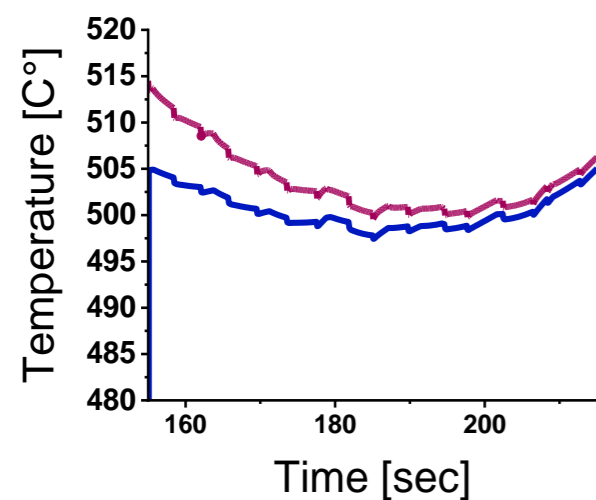
Selected results- Model predictions

MICROSTRUCTURE EVOLUTION

500°C, 0.01s⁻¹, ND



- Subgrains become larger
- Grains become smaller
- Homogenous grain/subgrain structure in P1 and P2

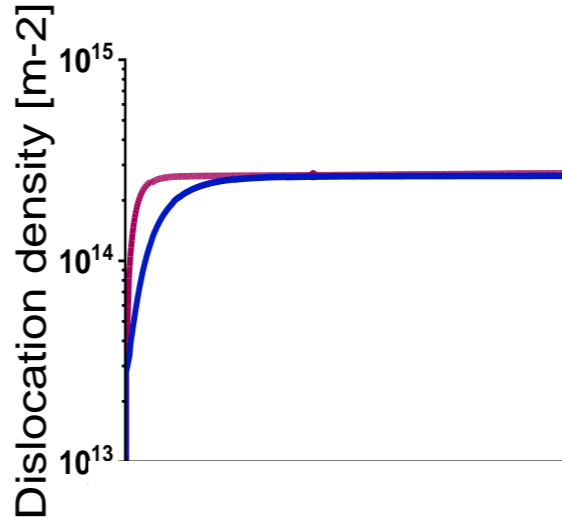
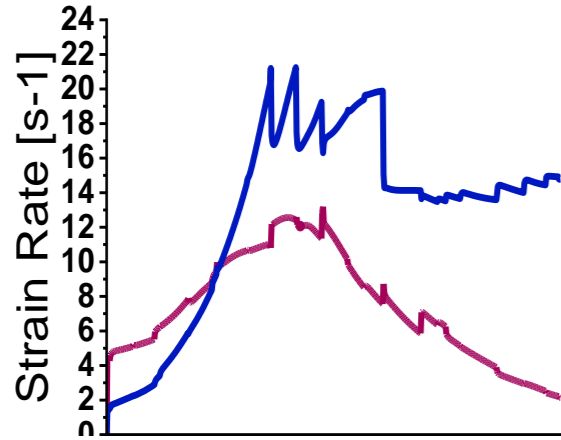


P1 ———
P2 ———

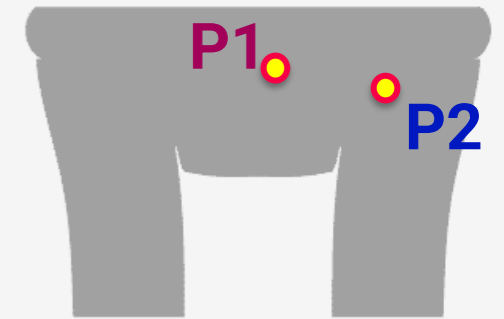
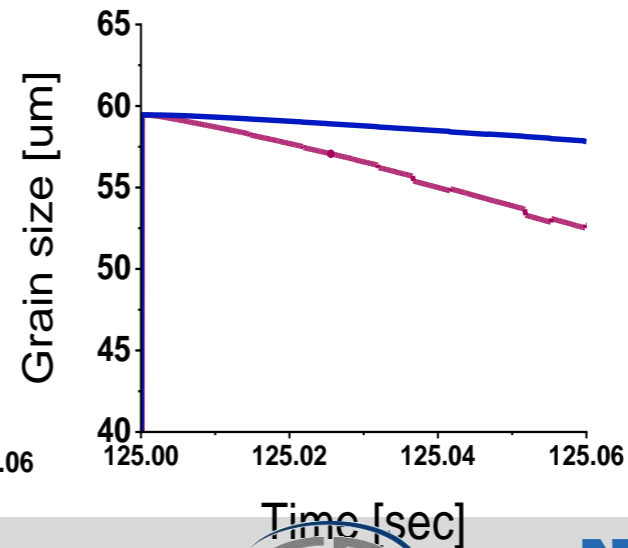
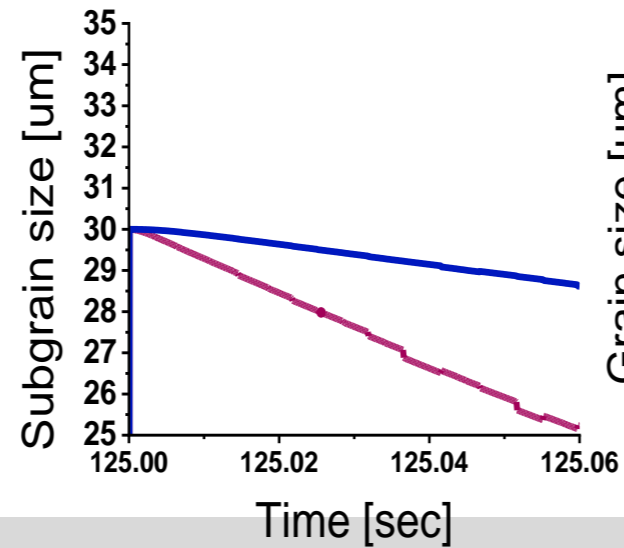
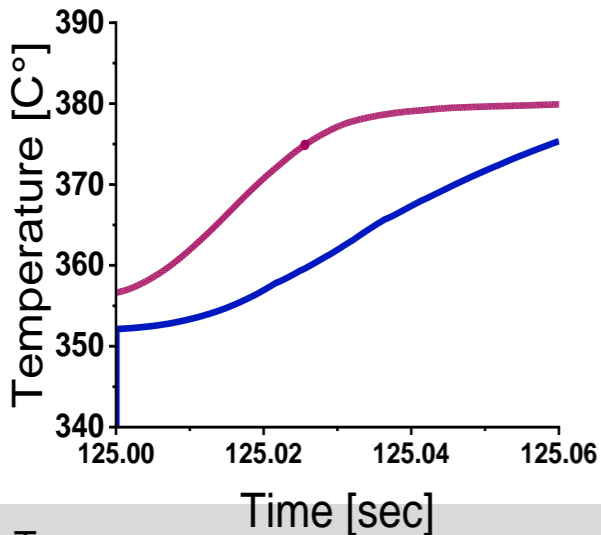
Selected results- Model predictions

MICROSTRUCTURE EVOLUTION

350°C, 10s⁻¹, ND



- Grains/subgrains become smaller
- Grains/subgrains smaller in P1 than P2

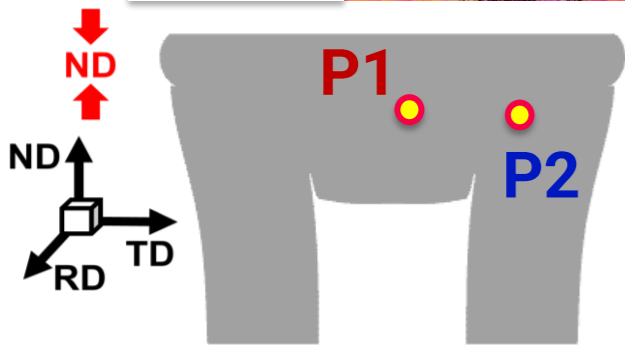
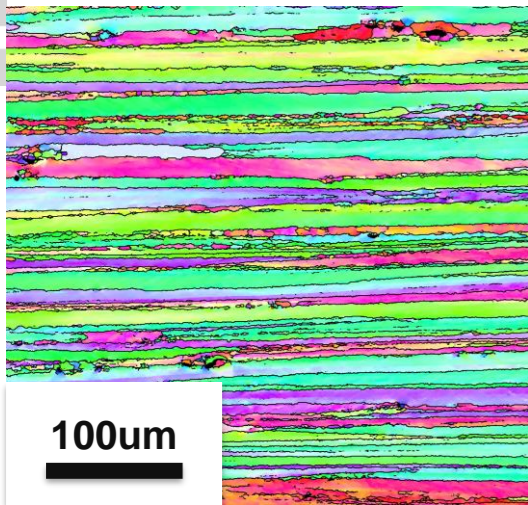


P1 ———
P2 ———

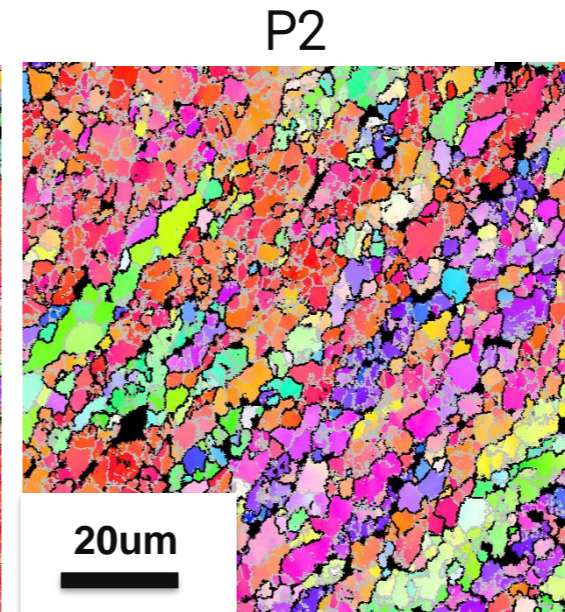
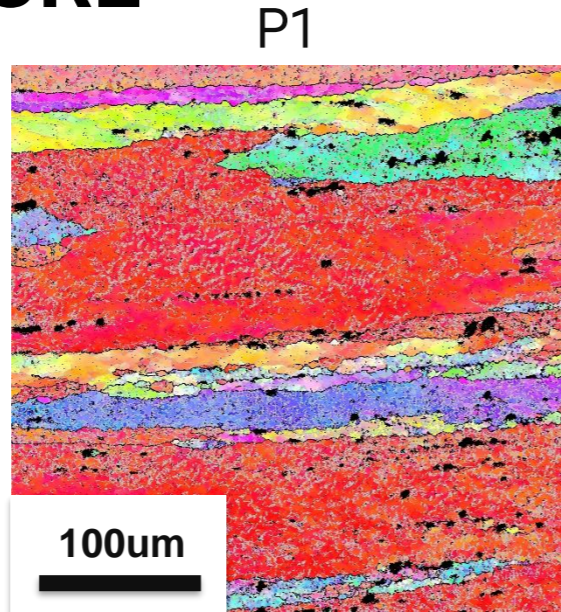
Selected results- Model predictions

MICROSTRUCTURE EVOLUTION

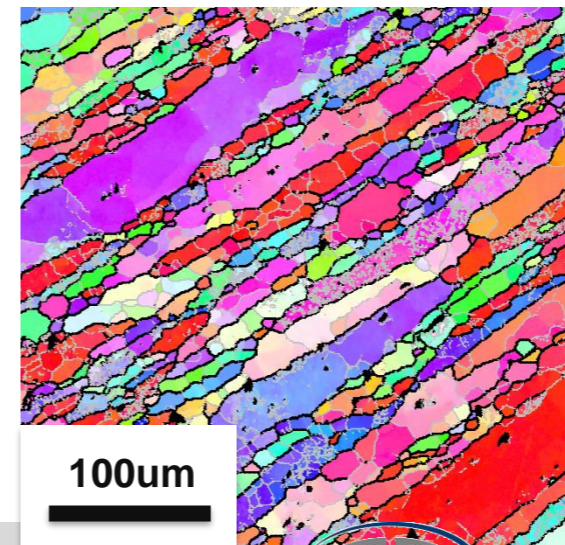
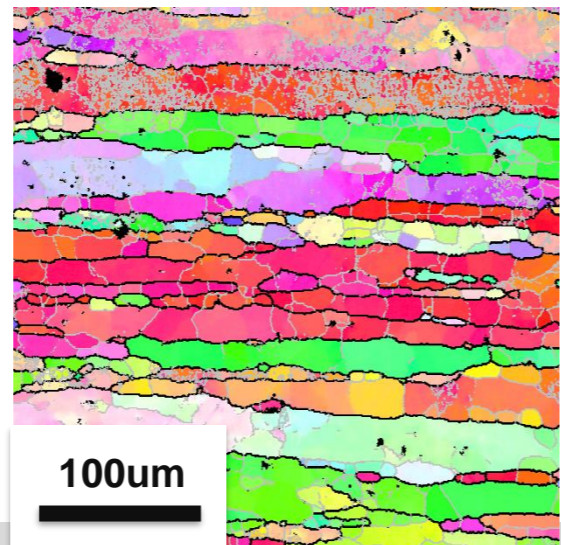
Base material



350°C, 10s⁻¹



500°C, 0.01s⁻¹

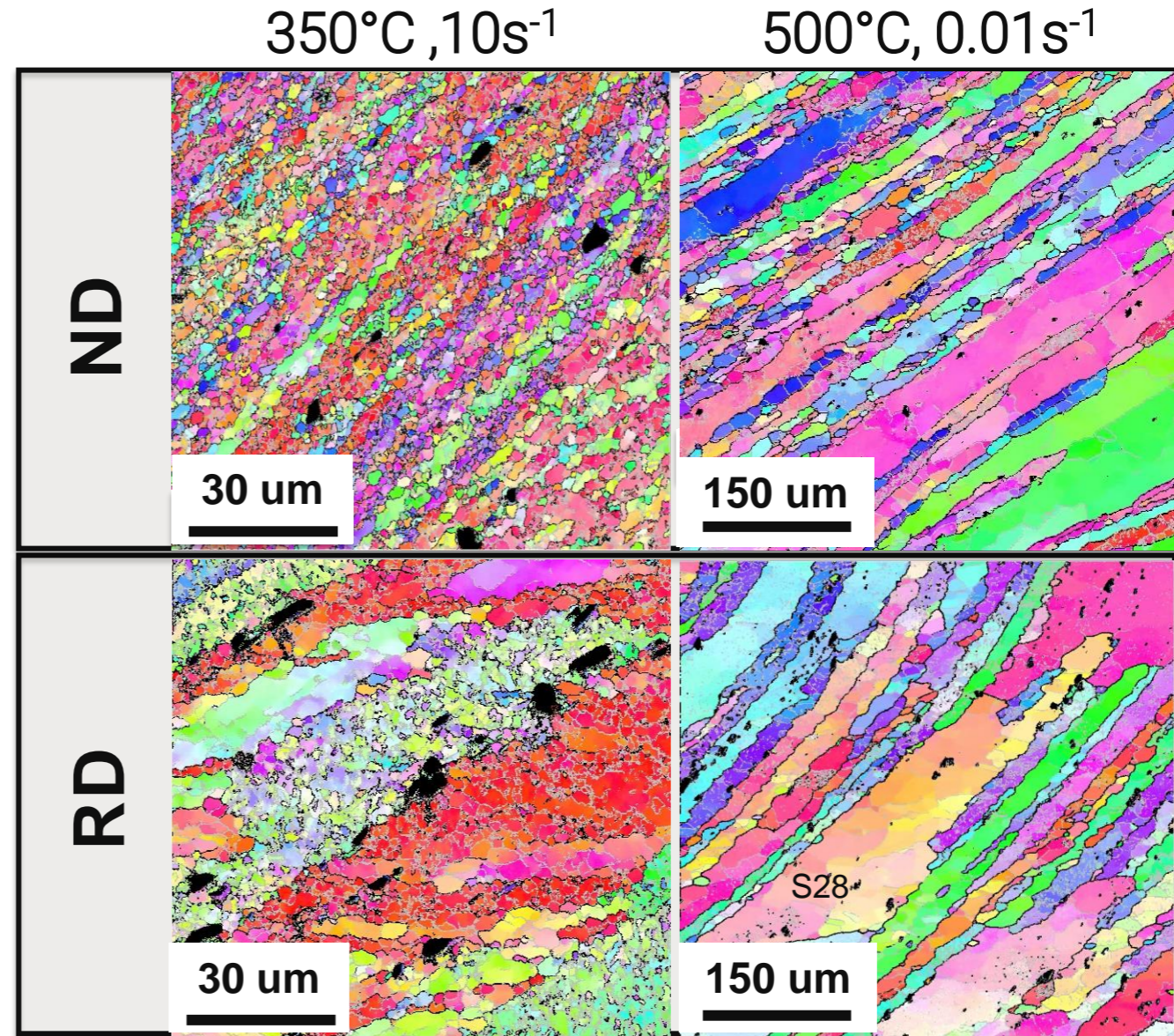
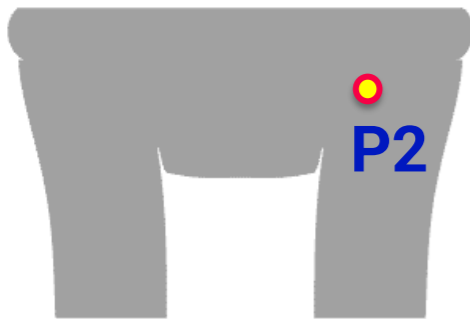


- Subgrains smaller in P1 than P2
- Grain sizes?

- Subgrains become larger than for 300°C 10s⁻¹
- Homogenous grain/subgrain structure in P1 and P2

Selected results

MICROSTRUCTURE EVOLUTION DURING SHEAR DEFORMATION



CONCLUSIONS

- Substructure becomes coarser for higher temperatures and lower strain rates
- For analyzed areas, higher temperatures and lower strain rates produce homogeneous grain and subgrain formation
- For low temperatures and high strain rates further analysis is required to assess grain/subgrain sizes
- Grain/subgrain formation is not influenced significantly by compression direction, but by the deformation conditions ($T, \dot{\epsilon}$)

SUMMARY AND ACKNOWLEDGEMENTS

- We generated heterogeneous deformation conditions using by hot compressing **hat-shaped samples** to represent **hot extrusion** processes
- The compression tests were modeled and simulated using a **physics-based model** implemented in **DEFORM** (FEM software)
- The **heterogeneous microstructural evolution** during hot deformation was studied

Funding

CHRISTIAN DOPPLER SOCIETY



NEUMAN ALUMINIUM



5th International Conference on
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Thank you for your attention!

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- [2] Ricardo Henrique Buzolin, Friedrich Krumphals & Maria Cecilia Poletti (2022): Topological aspects in the microstructural evolution of AA6082 during hot plastic deformation, European Journal of Materials, DOI: 10.1080/26889277.2022.2125350
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- [4] Ricardo Henrique Buzolin, Michael Lasnik, Alfred Krumphals, Maria Cecilia Poletti, A dislocation-based model for the microstructure evolution and the flow stress of a Ti5553 alloy, International Journal of Plasticity, Volume 136, 2021, 102862, ISSN 0749-6419, <https://doi.org/10.1016/j.ijplas.2020.102862>.

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